

# INFLUENCE OF ION IMPLANTATION ON SURFACE MAGNETIC PROPERTIES OF AMORPHOUS Fe—B RIBBONS<sup>1)</sup>

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The influence of  $N^+$  implantation in as-quenched and preannealed  $Fe_{83}B_{17}$  metallic glass was investigated by means of Kerr magnetooptical hysteresisgraph which gives information from an about 30—50 nm thick surface layer [1]. The implanted ions penetrate about 100—150 nm into the ribbon depending on the accelerating voltage. The coercive force of the surface layer depends significantly on the dose of the implanted  $N^+$ . The heat treatment before or after the  $N^+$  implantation contributes to the change of the coercive force too.

## I. INTRODUCTION

It has been recognized in recent years that the surface magnetic properties of metallic glasses play an important role in modifying the bulk magnetic properties [2, 3]. Surface crystallization and the stability of the amorphous phase after partial irradiation has also become a widely investigated subject. The work reported here was undertaken to examine the effect of  $N^+$  implantation on the coercive force.

## II. EXPERIMENTAL

All the samples we used were produced by the melt-spin method and had a thickness of about 40  $\mu m$ . The amorphism of the samples was checked by X-ray diffraction. In order to eliminate the quenched-in stresses they were preannealed at  $T = 150^\circ C$  for 0.5 h in an Ar protecting atmosphere. The first batch of samples was irradiated by different doses after the preannealing treatment. The second batch of samples was annealed at higher temperatures before the implantation of a dose  $\sigma = 10^{17} N^+ / cm^2$  with  $N^+$  ions in an  $N_2$  protecting

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atmosphere for 0.5 h. The third batch of samples was annealed in an Ar protecting atmosphere for 0.5 h after implantation. The ions to be implanted were accelerated up to a 100 keV energy and the temperature during implantation was not higher than  $T = 80^\circ\text{C}$ . The annealing temperature was always below the crystallization temperature of the  $\text{Fe}_3\text{B}_5$  alloy [5].

### III. RESULTS AND DISCUSSION

The  $H_c$  of the pre-annealed and  $N^+$  implanted samples remains constant up to a dose of about  $\sigma_0 = 4 \times 10^{16} \text{ N}^+/\text{cm}^2$  (Fig. 1). Above  $\sigma_0$  the coercive force increases significantly with the dose. According to X-ray and CEMS investigations  $\alpha$ -Fe crystals were formed in the surface of these high dose implanted  $\sigma > \sigma_0$  samples. There was no sign of a crystalline phase.

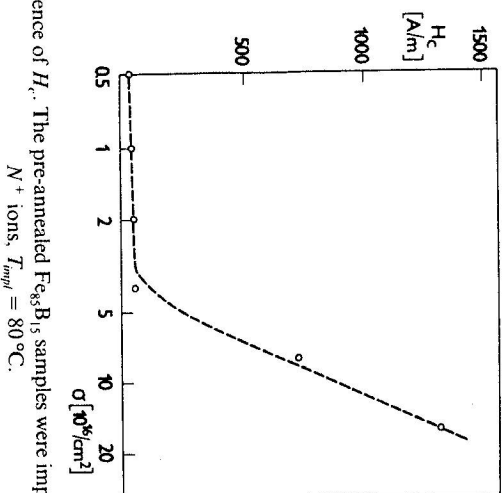


Fig. 1. Dose dependence of  $H_c$ . The pre-annealed  $\text{Fe}_3\text{B}_5$  samples were implanted with  $E = 100 \text{ keV}$   $N^+$  ions,  $T_{\text{impl}} = 80^\circ\text{C}$ .

The increase of the  $H_c$  is attributed to the appearance of  $\alpha$ -Fe microcrystals which inhibit the domain wall motion through magnetostriiction. It is believed that this crystallization is caused by the decrease of the B-atoms concentration due to their preferred sputtering during the implantation. A crystalline  $\text{Fe}_3\text{B}$  phase could not be formed because it is continuously amorphised by the implanted  $N^+$  ions.

Pre-annealings at higher temperatures cause also an increase in  $H_c$ , which is relatively small up to  $T = 390^\circ\text{C}$  (Fig. 2). Although the  $N^+$  implantation strongly enhanced the  $H_c$  of the sample surface layer, bulk investigation did not show any systematic difference between implanted and unimplanted samples.

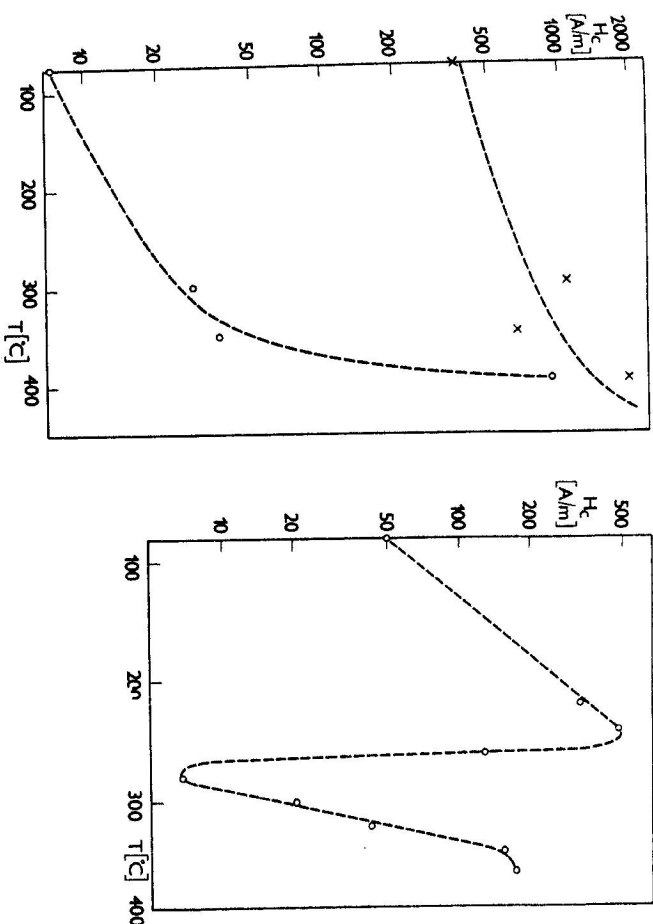


Fig. 2.  $\times$  The effect of pre-annealing in  $\text{N}_2$  protecting atmosphere on  $H_c$ ,  $t_{\text{annealing}} = 0.5 \text{ h}$ .  $\circ$  The effect of  $N^+$  implantation on pre-annealed samples.  $E = 100 \text{ keV}$ ,  $T_{\text{impl}} = 80^\circ\text{C}$ ,  $\sigma = 10^{17} \text{ N}^+/\text{cm}^2$ .

Fig. 3. The effect of annealing after implantation. The implantation parameters:  $E = 100 \text{ keV}$ ,  $T_{\text{impl}} = 80^\circ\text{C}$ ,  $\sigma = 4 \times 10^{16} \text{ N}^+/\text{cm}^2$ ,  $t_{\text{annealing}} = 0.5 \text{ h}$ .

The effect of annealing after implantation can be seen in Fig. 3. At low annealing temperatures  $T \lesssim 260^\circ\text{C}$  and increased  $H_c$  can be seen with a maximum at  $T \approx 250^\circ\text{C}$ . Between  $260^\circ\text{C}$ — $320^\circ\text{C}$   $H_c$  decreases with a minimum at  $T \approx 270^\circ\text{C}$ . Above  $T \approx 320^\circ\text{C}$  the heat treatments increase  $H_c$  again. It is assumed that the implantation destabilized surface [4] and the annealing results in the redundant iron crystallizing into  $\alpha$ -Fe and consequently in rising the  $H_c$ . At higher temperatures relaxation processes can also occur and they lead to the reduction of internal stresses.

The rising of  $H_c$  at temperatures  $T > 270^\circ\text{C}$  is attributed to pre-crystallization and to the influence of implantation.

### IV. CONCLUSIONS

The increasing of  $H_c$  as an effect of  $N^+$  implantation is attributed to  $\alpha$ -Fe microcrystals formed in the surface above a certain dose. The implantation enhanced  $H_c$  can be effectively reduced by post-annealing treatments.

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## ВЛИЯНИЕ ИОННОЙ ИМПЛАНТАЦИИ НА ПОВЕРХНОСТНЫЕ МАГНИТНЫЕ СВОЙСТВА АМОРФНЫХ Fe—В ЛЕНТ

При использовании керровского магнитооптического гистерезиграфа [1] было исследовано влияние  $N^+$  имплантации на  $Fe_8V_2$  магнитические стекла. Имплантированные ионы проникают только на глубину 100—150 нм в ленту, глубина проникновения зависит от ускоряющего напряжения. Поэтому влияние на коэрцитивную силу, обусловленное ионами, может быть изучено более точно, чем в объемных исследованиях. Наши исследования дают информацию из поверхностного слоя толщиной 30—50 нм.